



# Holey structured graphitic carbon nitride thin sheets with edge oxygen doping via photo-Fenton reaction with enhanced photocatalytic activity

ShuaiNan Guo, Yong Zhu, YunYun Yan, YuLin Min\*, JinChen Fan\*, QunJie Xu\*

Shanghai Key Laboratory of Materials Protection and Advanced Materials in Electric Power, Shanghai University of Electric Power, 200090, China

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## ABSTRACT

Holey structured graphitic carbon nitride thin sheets with edge oxygen doping (HS g-C<sub>3</sub>N<sub>4</sub>-O) via photo-Fenton reaction were reported. The HS g-C<sub>3</sub>N<sub>4</sub>-O exhibited high surface area of 348 m<sup>2</sup> g<sup>-1</sup> and narrow bandgap (2.434 eV), hence producing high photocatalytic activities for both H<sub>2</sub> generation and Rh.B degradation under solar light irradiation.

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## 1. Introduction

Photocatalytic hydrogen generation from water using photo-driven reaction has attracted considerable attention as an ideal green technology [1–3]. Of these, a stable photocatalysts which high effectively converts solar energy into chemical fuels is an appreciated alternative. Among various semiconductors photocatalyst, metal-free graphitic carbon nitride (g-C<sub>3</sub>N<sub>4</sub>) with a suitable band gap and edges has emerged as a new class of photocatalyst due to its nontoxicity, thermal and chemical stability. However, the latest documented quantum yields for H<sub>2</sub> production using pristine g-C<sub>3</sub>N<sub>4</sub> does not exceed 4% [4–7], which is far from satisfactory for further application.

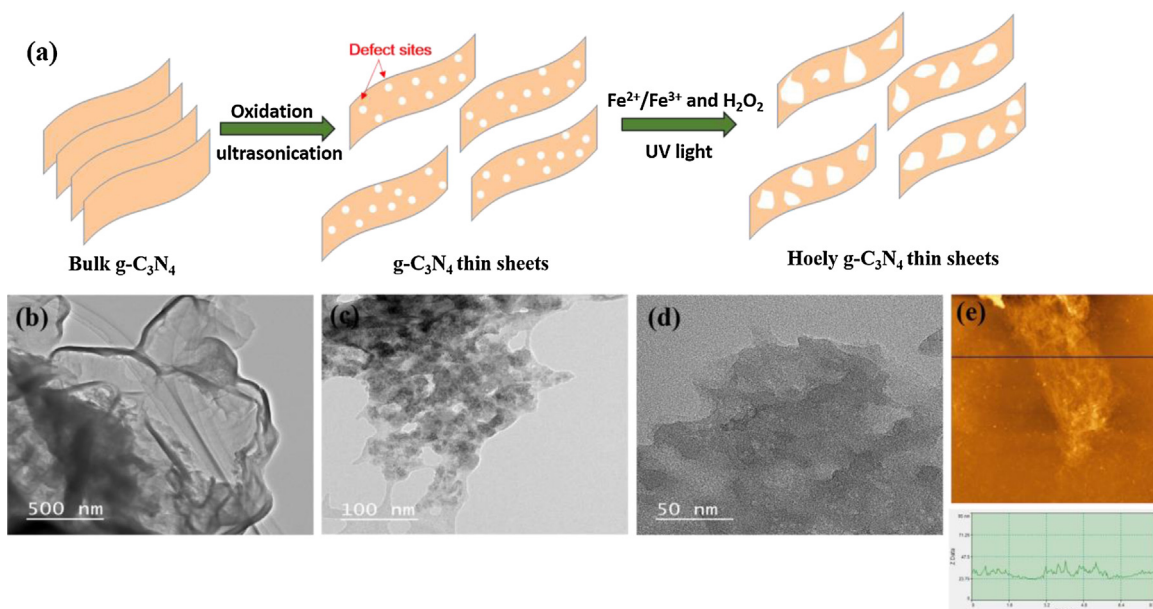
As is known, an excellent photocatalyst must possess a high specific surface area to take place redox reactions and a minimizing recombination to take advantage photo-induced charge. Undeniably, the bulk g-C<sub>3</sub>N<sub>4</sub> has poor surface area and the high recombination rate. To improve the essential properties of g-C<sub>3</sub>N<sub>4</sub>, various strategies including heteroatom doping [8–11], layer exfoliation [12], heterojunction configuration [12–15] and carbonaceous scaffold [16–18] have been adopted. Meanwhile, Oxygen

heteroatom has been successfully introduced by a simple H<sub>2</sub>O<sub>2</sub> oxidation and its photoactivity has been improved to some extent [19,20]. However, the strongly oxidizing agent of H<sub>2</sub>O<sub>2</sub> would be easily break the g-C<sub>3</sub>N<sub>4</sub> structure whose is possibly converted into carbon or nitrite oxides. In this sense, rationally controlling the oxidation process would leave behind many vacancies, which could be gradually extended into nanopores texture. The expected oxygen doped g-C<sub>3</sub>N<sub>4</sub> with high porosity is able to exhibit impressive photocatalytic activity due to simultaneous doping effect and high active sites.

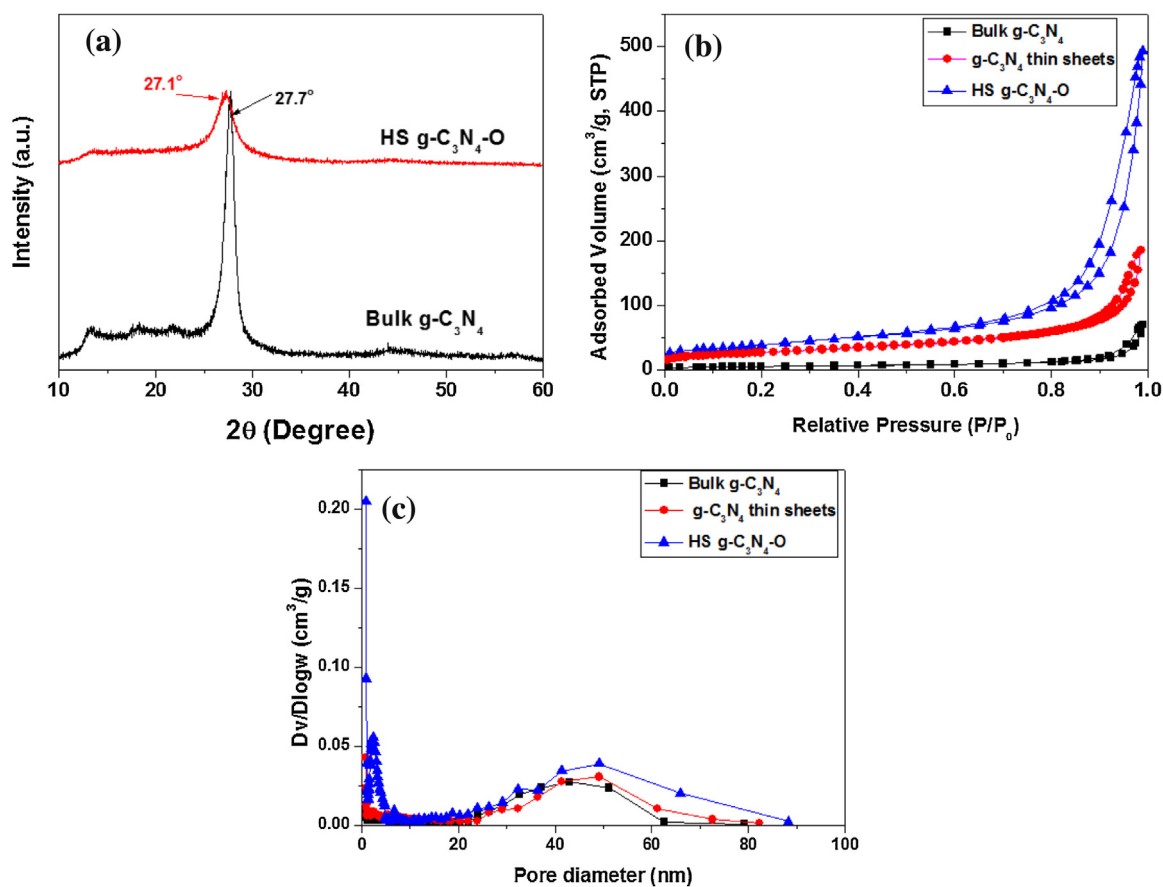
In this study, a holey structured g-C<sub>3</sub>N<sub>4</sub> with edge oxygen doping (HS g-C<sub>3</sub>N<sub>4</sub>-O) was successfully obtained using photo-Fenton reaction in presence of Fe<sup>3+</sup>/Fe<sup>2+</sup> and H<sub>2</sub>O<sub>2</sub>. First, bulk g-C<sub>3</sub>N<sub>4</sub> was oxidized and exfoliated into g-C<sub>3</sub>N<sub>4</sub> thin sheets with abundant oxygen-containing groups, which based on Niu et al. report [12]. Furthermore, it has also been demonstrated that the photo-Fenton reaction can generate hydroxyl radicals (OH•) under the photoassisted catalysis of Fe<sup>3+</sup>/Fe<sup>2+</sup> in present of H<sub>2</sub>O<sub>2</sub>, and which has been considered as one of the most powerful oxidizing species [21]. Thus, the g-C<sub>3</sub>N<sub>4</sub> thin sheets with abundant oxygen-containing groups as active defective sites can be partially oxidized and etched, leaving behind carbon vacancies, which gradually extend into nanopores [22,23]. The as-resulted HS g-C<sub>3</sub>N<sub>4</sub>-O demonstrated a narrow bandgap of 2.434 eV and high surface area of 348 m<sup>2</sup> g<sup>-1</sup>, thus presenting highly efficient photocatalytic activity for H<sub>2</sub> generation and Rh.B degradation.

\* Corresponding authors.

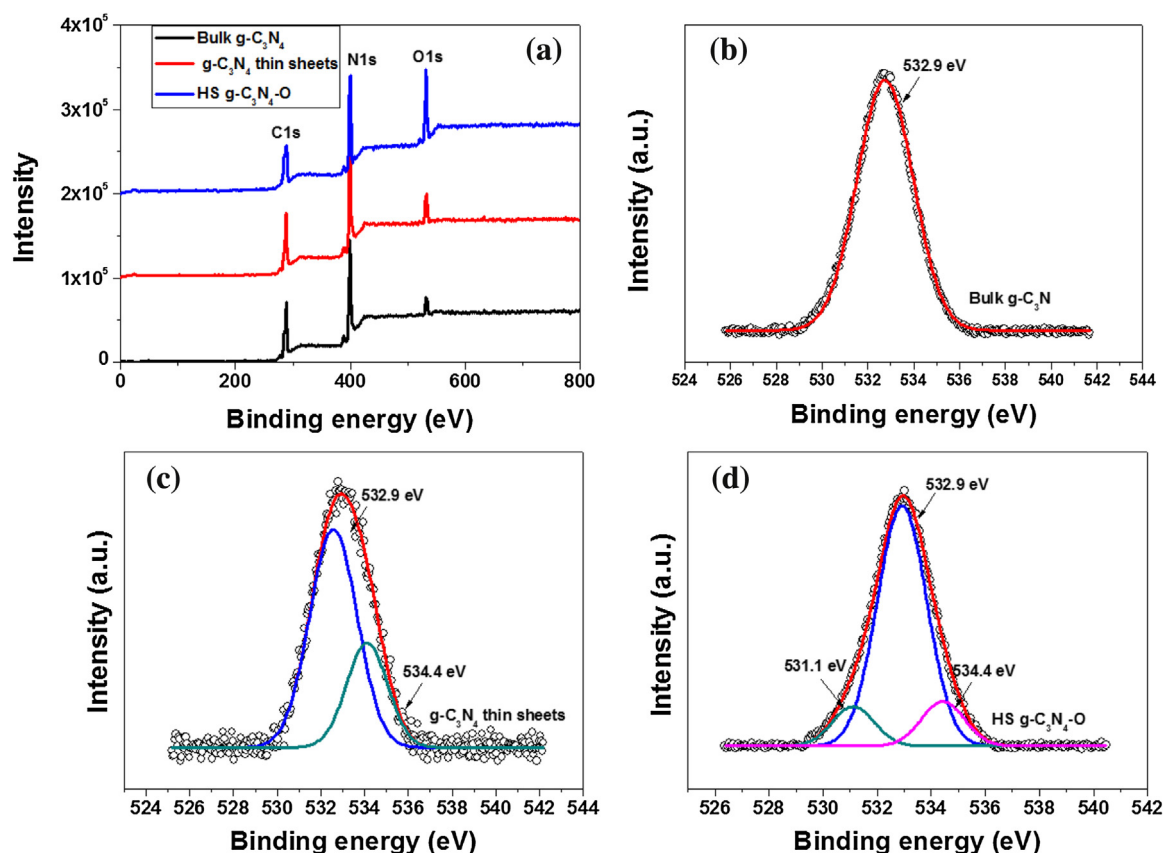
E-mail addresses: [ahaqmylin@126.com](mailto:ahaqmylin@126.com) (Y. Min), [jinchen.fan@gmail.com](mailto:jinchen.fan@gmail.com) (J. Fan), [xuqunjie@shiep.edu.cn](mailto:xuqunjie@shiep.edu.cn) (Q. Xu).



**Fig. 1.** (a) Schematic illustration of the HS g-C<sub>3</sub>N<sub>4</sub>-O formation via photo-Fenton reaction. (b) TEM image of oxidized g-C<sub>3</sub>N<sub>4</sub> thin sheets. (c and d) TEM image of the HS g-C<sub>3</sub>N<sub>4</sub>-O. (e) AFM image of the HS g-C<sub>3</sub>N<sub>4</sub>-O.



**Fig. 2.** (a) XRD patterns of the bulk g-C<sub>3</sub>N<sub>4</sub> and HS g-C<sub>3</sub>N<sub>4</sub>-O. (b) Nitrogen adsorption-desorption isotherms and the corresponding pore size distribution curves (c) of the bulk g-C<sub>3</sub>N<sub>4</sub>, g-C<sub>3</sub>N<sub>4</sub> thin sheets and HS g-C<sub>3</sub>N<sub>4</sub>-O.



**Fig. 3.** (a) Full XPS spectra of the bulk  $g\text{-C}_3\text{N}_4$ ,  $g\text{-C}_3\text{N}_4$  thin sheets and HS  $g\text{-C}_3\text{N}_4\text{-O}$ . (b) High resolution O1s XPS spectra of the bulk  $g\text{-C}_3\text{N}_4$ . (c) High resolution O1s XPS spectra of the  $g\text{-C}_3\text{N}_4$  thin sheets. (d) High resolution O1s XPS spectra of HS  $g\text{-C}_3\text{N}_4\text{-O}$ .

## 2. Experimental

### 2.1. Materials

Rhodamine B,  $\text{H}_2\text{SO}_4$ ,  $\text{HNO}_3$ ,  $\text{H}_2\text{O}_2$  were purchased from Shanghai Pure Chemical Co., Ltd., China. Other materials were obtained from Sigma-Aldrich.

### 2.2. Synthesis of HS $g\text{-C}_3\text{N}_4\text{-O}$

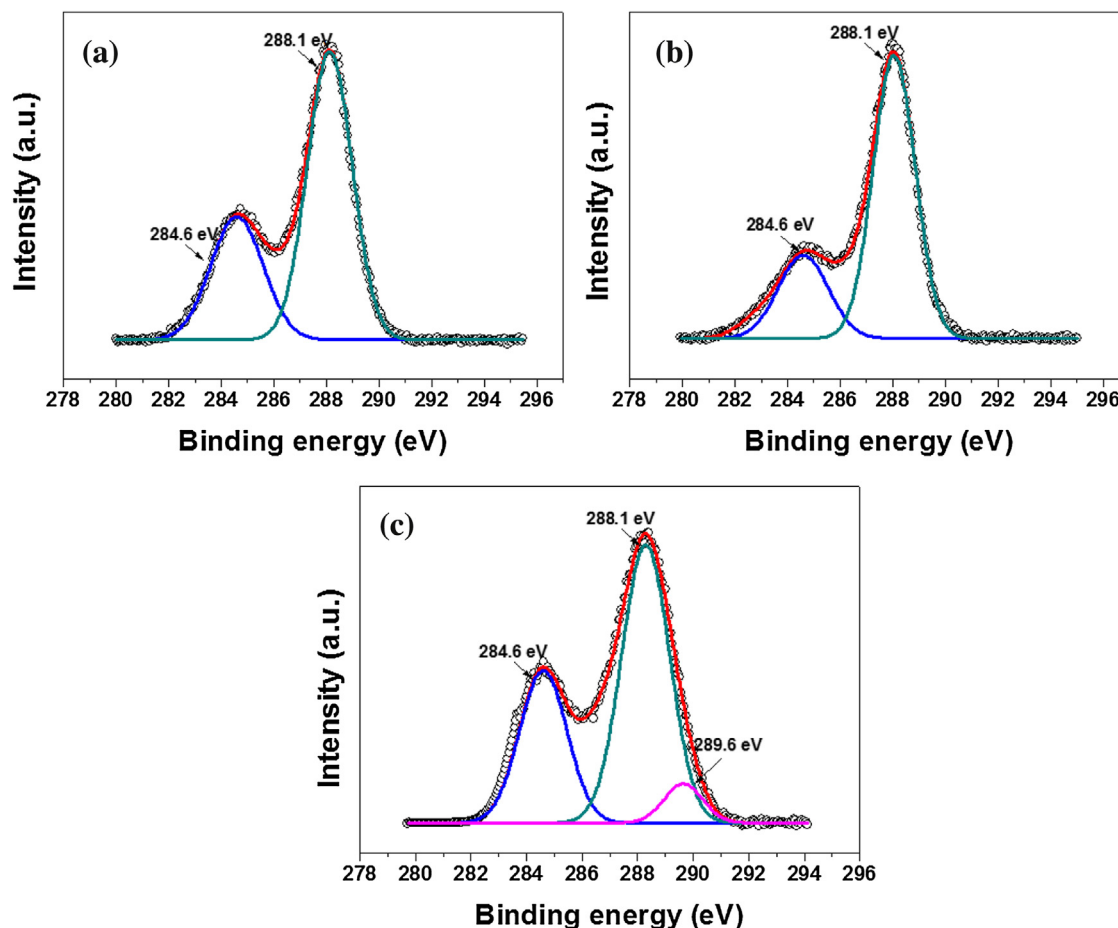
Bulk  $g\text{-C}_3\text{N}_4$  was synthesized by directly heating melamine in the semiclosed system to prevent the sublimation of melamine at  $550^\circ\text{C}$  for 4 h. First, 1 g of bulk  $g\text{-C}_3\text{N}_4$  was treated in the mixture of concentrated  $\text{H}_2\text{SO}_4$  and  $\text{HNO}_3$  with 1:1 volume ratio for about 2 h at room temperature. The mixture was then diluted with deionized water (1 L) and washed for several times, the as-obtained white product was  $g\text{-C}_3\text{N}_4$  thin sheets. The photo-Fenton reactions of  $g\text{-C}_3\text{N}_4$  thin sheets were carried out under UV light irradiation ( $\lambda = 365\text{ nm}$ ). The reaction is carried out under vigorous stirring in a 40 mL quartz tube which is installed 5 cm away from the irradiation lamp. In a typical experiment, 0.5 mg  $g\text{-C}_3\text{N}_4$  thin sheets were dispersed in 5 mL water, then, 20 mL of 20 mM  $\text{H}_2\text{O}_2$ , and 100  $\mu\text{L}$  of 1.0 mM of  $\text{FeCl}_3$  were added. The pH of the mixture was adjusted to 4. The reaction was maintained around 5 h under continuous irradiation of UV light. The reaction products were washed with 0.1 mmol HCl and deionized water several time to remove residual iron ions until the pH 7. The HS  $g\text{-C}_3\text{N}_4\text{-O}$  was collected by centrifugation and dried at  $60^\circ\text{C}$ .

### 2.3. Analysis instruments

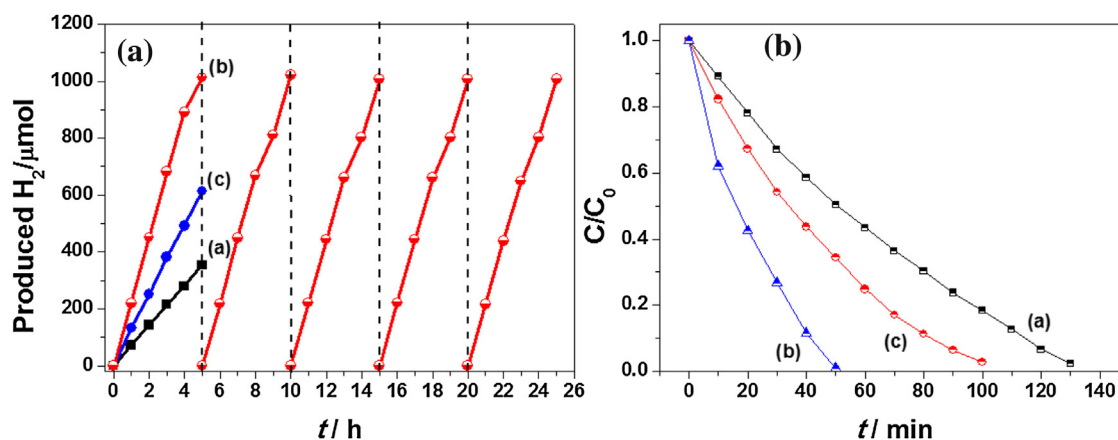
The  $\text{N}_2$  adsorption isotherm was measured at 77 K using a BEL sorp Analyzer (BEL, Japan). The XRD patterns were obtained with a diffractometer on Rigaku, Japan, RINT 2500V using  $\text{Cu K}\alpha$  radiation. The Fourier transform-infrared (FT-IR) spectra were recorded in KBr pellets with a Bruker FT-IR. The diffuse reflectance UV-vis absorption spectra were obtained using a spectrophotometer (Shimadzu UV-2401PC) equipped with a diffuse reflectance accessory, and  $\text{BaSO}_4$  was used as the reference. Transmission electron microscopy (TEM) observations were obtained using a JEM-2200FS microscope with Cs correction. The photoluminescence of the powder samples was recorded at room temperature using a fluorescence spectrometer (Shimadzu UV-2401PC). The UV-vis absorption spectra were recorded using a (S-3100, Sainco. Co., Ltd.) spectrometer.

### 2.4. Photoelectrochemical measurements

Photocurrent responses were carried out in a conventional three-electrode system using a potentiostat (CH Instruments, CHI 660). The photocurrents were collected and measured via electron shuttles on an inert Pt electrode immersed in the 0.5 M  $\text{Na}_2\text{SO}_4$  solution at 0.2 V potential bias under a solar simulator with Xe lamp through a UV-cutoff filter ( $>420\text{ nm}$ ) served as a light source. The electrodes was prepared by dropping 20  $\mu\text{L}$  aqueous slurry that consisted of 50 mg of prepared sample with 200  $\mu\text{L}$  of  $\text{H}_2\text{O}$ , 5  $\mu\text{L}$  of acetylacetone, and 5  $\mu\text{L}$  of Nafion on a cleaned ITO glass substrate. After air dried, the electrode was annealed at  $300^\circ\text{C}$  for



**Fig. 4.** (a) High resolution C1s XPS spectra of the bulk g-C<sub>3</sub>N<sub>4</sub>. (b) High resolution C1s XPS spectra of the g-C<sub>3</sub>N<sub>4</sub> thin sheets. (c) High resolution C1s XPS spectra of HS g-C<sub>3</sub>N<sub>4</sub>-O.



**Fig. 5.** (a) Photocatalytic activities toward H<sub>2</sub> production of (a) bulk g-C<sub>3</sub>N<sub>4</sub>, (b) HS g-C<sub>3</sub>N<sub>4</sub>-O and (c) g-C<sub>3</sub>N<sub>4</sub> thin sheets under visible light irradiation. (b) Photocatalytic activities toward Rh.B decolorization of (a) bulk g-C<sub>3</sub>N<sub>4</sub>, (b) HS g-C<sub>3</sub>N<sub>4</sub>-O and (c) g-C<sub>3</sub>N<sub>4</sub> thin sheets under visible light irradiation.

2 h in air. Mott–Schottky tests was performed with the potentials ranged from  $-0.5$  V to  $1$  V (vs Ag/AgCl) at the selected frequencies of  $1.5$  kHz.

### 2.5. Photocatalytic test

For photodegradation of organic dyes, the decomposition reaction of a  $50$  mL Rh.B ( $2 \times 10^{-5}$  M) aqueous solution was carried out. A powdered sample of  $30$  mg was dispersed in the Rh.B

solution under ultrasonication for  $1$  min, and then was kept in dark conditions for  $1.5$  h. For the irradiation system, a  $300$  W Xe lamp with  $420$  nm filter was used at the distance of  $100$  mm from the solution in a darkness box. The suspension was irradiated for different irradiation times. The samples were then withdrawn regularly from the reactor and the dispersed powder was removed through centrifugation. The clean transparent solution was analyzed by UV–vis spectroscopy. The concentration of Rh.B in the solution was determined as a function of irradiation time from the absorbance



**Table 1**

BET surface area, total pore volume and average pore size of bulk g-C<sub>3</sub>N<sub>4</sub>, g-C<sub>3</sub>N<sub>4</sub> thin sheet and HS g-C<sub>3</sub>N<sub>4</sub>-O.

Item	Bulk g-C <sub>3</sub> N <sub>4</sub>	g-C <sub>3</sub> N <sub>4</sub> thin sheet	HS g-C <sub>3</sub> N <sub>4</sub> -O
$S_{\text{BET}}$ (m <sup>2</sup> g <sup>-1</sup> )	46	106	348
$V_{\text{total}}$ (cm <sup>3</sup> g <sup>-1</sup> )	0.041	0.075	0.155
$D$ (nm)	12.8	13.2	17.6

region at a functional wavelength. Photocatalytic hydrogen evolution was carried out in a Pyrex top-irradiation reaction vessel connected to a closed glass gas system. In detail, 30 mg of catalyst powders was dispersed in the aqueous solution (200 mL) containing triethanolamine (20 vol.%) as sacrificial agent. Co-catalysts Pt nanoparticles were introduced by in-situ photodeposition method as reported before, where loaded amount of Pt was estimated to be about 5 wt.%. 1 mL sample of the generated gas was collected using a glass injection syringe which could be sealed up and the hydrogen and oxygen contents were analyzed using a gas chromatograph.

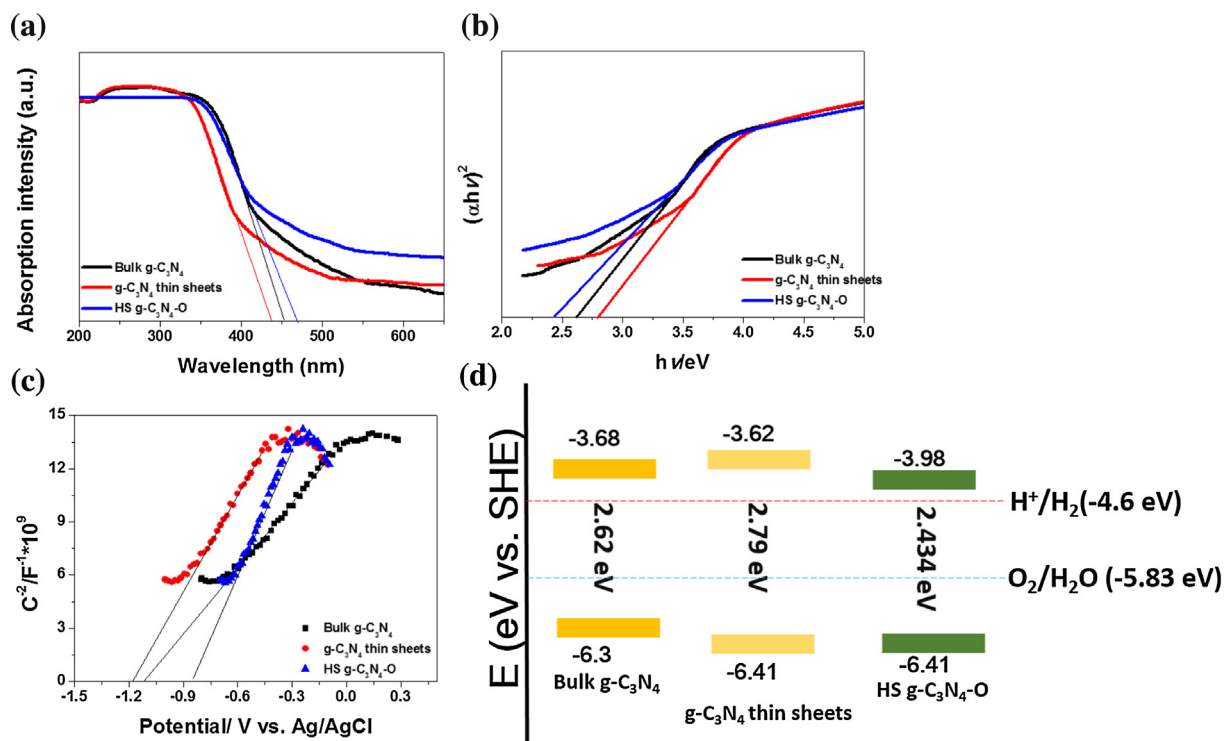
### 3. Results and discussion

The formation processes of HS g-C<sub>3</sub>N<sub>4</sub>-O is schematically shown in Fig. 1a. The SEM images of the bulk g-C<sub>3</sub>N<sub>4</sub> and oxidized g-C<sub>3</sub>N<sub>4</sub> thin sheets are shown in Fig. S1. The TEM image of oxidized g-C<sub>3</sub>N<sub>4</sub> thin sheets was exhibited in Fig. 1b, demonstrating a surface rough sheet with large area. Furthermore, the oxidized g-C<sub>3</sub>N<sub>4</sub> thin sheets have a tendency to edge bend due to minimizing the Van der Waals' force and surface energy [12]. Fig. 1c and d indicates that the etching process could occur across the entire g-C<sub>3</sub>N<sub>4</sub> thin sheets to result in abundant in-plane pores of a few nanometers, demonstrating holey structured g-C<sub>3</sub>N<sub>4</sub> thin sheets. AFM image of the holey structured g-C<sub>3</sub>N<sub>4</sub> thin sheets appears as loose and soft agglomerates with a size of tens of micrometers (Fig. 1e). The compared with the pristine g-C<sub>3</sub>N<sub>4</sub>, the crystallinities of HS g-C<sub>3</sub>N<sub>4</sub>-O are weakened at  $\sim 13^\circ$ , corresponding to a weak interplanar structure. The (002) peak of HS g-C<sub>3</sub>N<sub>4</sub>-O slightly shifted toward low angle, indicating a more loose of C–N planes due to oxygen doping (Fig. 2a). The effects of holey structure on the BET surface area and porous structures of g-C<sub>3</sub>N<sub>4</sub> thin sheets were investigated using adsorption–desorption measurements. As shown in Fig. 2b, the remarkable hysteresis loops indicate the mesoporous nature (2–50 nm) of these g-C<sub>3</sub>N<sub>4</sub> and their derivatives. Moreover, the adsorption branch of nitrogen isotherms shows a steady increase at  $P/P_0$  approaching unity, suggesting the formation of large mesopores and small macropores. The pore size distributions of bulk g-C<sub>3</sub>N<sub>4</sub>, g-C<sub>3</sub>N<sub>4</sub> thin sheets and HS g-C<sub>3</sub>N<sub>4</sub>-O are shown in Fig. 2c. The pore-size distributions are very broad, indicating the existence of mesopores and macropores. Particularly, the HS g-C<sub>3</sub>N<sub>4</sub>-O shows a more prominent pore size distribution in the range of 1–5 nm, which could be ascribed to the nanopores in the basal plane of HS g-C<sub>3</sub>N<sub>4</sub>-O, which is consistent with TEM analysis. The porous properties of these materials are summarized in Table 1, including the specific surface area ( $S_{\text{BET}}$ ), total pore volume, and average pore diameter. It is clearly seen that the  $S_{\text{BET}}$  of both g-C<sub>3</sub>N<sub>4</sub> thin sheets (106 m<sup>2</sup> g<sup>-1</sup>) and HS g-C<sub>3</sub>N<sub>4</sub>-O (348 m<sup>2</sup> g<sup>-1</sup>) is higher than that of bulk g-C<sub>3</sub>N<sub>4</sub> (46 m<sup>2</sup> g<sup>-1</sup>), respectively. A greater specific surface area of photocatalysts can supply more surface active sites and make charge carriers transport easier, leading to an enhancement of the photocatalytic performance [24].

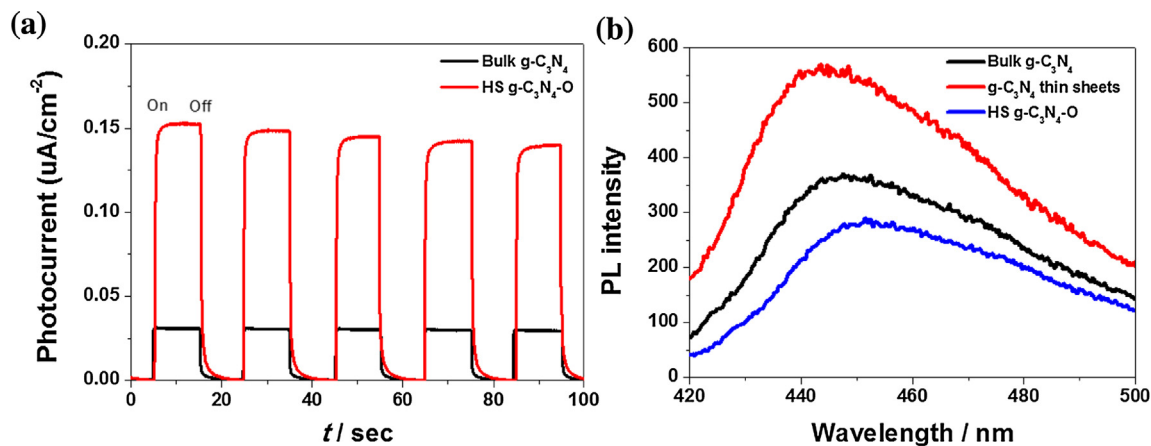
X-ray photoelectron spectroscopy (XPS) was employed to confirm oxygen doping species in HS g-C<sub>3</sub>N<sub>4</sub>-O. Besides the appearance of expected C and N elements, comparison full XPS spectra among bulk g-C<sub>3</sub>N<sub>4</sub>, g-C<sub>3</sub>N<sub>4</sub> thin sheets and HS g-C<sub>3</sub>N<sub>4</sub>-O clearly reveals that significant content of oxygen is doped in HS g-C<sub>3</sub>N<sub>4</sub>-O (Fig. 3a). Their high-resolution O1s XPS spectra show core level at 532.9 eV, which is ascribed to adsorbed water (Fig. 3b–d) [25]. It is

understandable the intensity increase in HS g-C<sub>3</sub>N<sub>4</sub>-O because the HS g-C<sub>3</sub>N<sub>4</sub>-O has much high surface area, leading to numerous water adsorbed on surface. Furthermore, the deconvolution of O1s XPS exhibited two additional peaks at 531.1 and 534.4 eV in the HS g-C<sub>3</sub>N<sub>4</sub>-O, which correspond to N–C–O bond and adsorbed O<sub>2</sub>, respectively [19,26]. However, the N–C–O bond at 531.1 eV could not be detected in g-C<sub>3</sub>N<sub>4</sub> thin sheets, which indicates the oxygen doping mainly occurred during photo-Fenton reaction and located at the g-C<sub>3</sub>N<sub>4</sub> edge, which is consistent with FT-IR analysis (Fig. S2). To further eliminate possible other kinds of bonds, such as carbon vacancies, their deconvolutions of C1s XPS are shown in Fig. 4. It can be clearly observed there is two main peaks located at 284.6 and 288.1 eV, corresponding to characteristic CC and C–N–C bonds in g-C<sub>3</sub>N<sub>4</sub>. However, a distinctive peak located at 289.6 eV is found in the HS g-C<sub>3</sub>N<sub>4</sub>-O (Fig. 4c), which could be ascribed to oxygen atom substitute for nitrogen atom because of easy migration of electrons bound to oxygen.

The photocatalytic H<sub>2</sub> evolution over bulk g-C<sub>3</sub>N<sub>4</sub>, HS g-C<sub>3</sub>N<sub>4</sub>-O and g-C<sub>3</sub>N<sub>4</sub> thin sheets was tested under simulated solar light using triethanolamine as sacrificial agent. As shown in Fig. 5a, the H<sub>2</sub> evolution rate of the bulk g-C<sub>3</sub>N<sub>4</sub> is 70.65  $\mu\text{mol/h}$ , while that of g-C<sub>3</sub>N<sub>4</sub> thin sheets achieve to 122.56  $\mu\text{mol/h}$ . It is believed that the increased H<sub>2</sub> evolution rate are associated with the improved electron transport and/or electronic band structure changes induced by quantum confinement effect in the thin sheets [12]. When further increased the surface area and doped with oxygen in HS g-C<sub>3</sub>N<sub>4</sub>-O, the H<sub>2</sub> evolution rate of the bulk g-C<sub>3</sub>N<sub>4</sub> dramatically increase to 202.56  $\mu\text{mol/h}$  and the evolution rate is quite stable under continuous irradiation of 26 h. The enhanced photocatalytic performance is further evaluated by decolorizing Rh.B under simulated solar light. As shown in Fig. 5b, complete decolorization of Rh.B over HS g-C<sub>3</sub>N<sub>4</sub>-O could be carried out within 50 min. However, the complete decolorizations of Rh.B over bulk g-C<sub>3</sub>N<sub>4</sub> and g-C<sub>3</sub>N<sub>4</sub> thin sheets need take 130 and 100 min under the same condition as the process for HS g-C<sub>3</sub>N<sub>4</sub>-O, respectively. These results demonstrate a significant enhancement in both photocatalytic H<sub>2</sub> generation and decolorizations of Rh.B using HS g-C<sub>3</sub>N<sub>4</sub>-O, the mechanism of which will be explained from the analysis of Fig. 6. As known, the semiconductor photocatalysis process is composed of the photoexcitation, the separation and transfer of photoinduced charge carriers, and the subsequent photocatalytic reaction. The photoexcitation primarily depends on the bandgap structure of the semiconductor. The UV–vis absorption spectra of bulk g-C<sub>3</sub>N<sub>4</sub>, g-C<sub>3</sub>N<sub>4</sub> thin sheets and HS g-C<sub>3</sub>N<sub>4</sub>-O are shown in Fig. 6a. The band gap of photocatalysts could be determined by using the equation:  $(\alpha h\nu)^n = A(h\nu - E_g)$ , according to Fig. 6a, where  $\alpha$  is the absorption coefficient,  $h\nu$  is the light energy,  $A$  is a constant,  $E_g$  is the optical band gap energy, and  $n$  is equal to 1/2 for an indirect band gap (Fig. 6b). It is estimated that the indirect band gaps of the bulk g-C<sub>3</sub>N<sub>4</sub>, g-C<sub>3</sub>N<sub>4</sub> thin sheets and HS g-C<sub>3</sub>N<sub>4</sub>-O are 2.62, 2.79 and 2.434 eV, respectively. It can be obviously seen that the band gap of g-C<sub>3</sub>N<sub>4</sub> thin sheets is larger than bulk g-C<sub>3</sub>N<sub>4</sub> due to the well-known quantum confinement effect by shifting the conduction and valence band edges in opposite directions [27]. After edge doping with oxygen, the band gap of HS g-C<sub>3</sub>N<sub>4</sub>-O is significantly reduced to 2.434 eV, suggesting more visible light harvesting capability. The result is reasonable because an oxygen atom owns one more electron than an N atom does, naturally, the substitution of N sites by O atoms would offer extra electrons in the HS g-C<sub>3</sub>N<sub>4</sub>-O in analogy to the Nb-doped TiO<sub>2</sub> system [28]. Subsequently, the extra electrons would be redistributed to their nearest carbon atoms and delocalized among the big  $\pi$  bonds of the g-C<sub>3</sub>N<sub>4</sub>, resulting in the formation of a defect-related state below the conduction band maximum (CBM) of the g-C<sub>3</sub>N<sub>4</sub> [19]. The CBM could be estimated by Mott–Schottky plots, as shown in Fig. 6c. The flat band potentials ( $V_{\text{fb}}$ ) of bulk g-C<sub>3</sub>N<sub>4</sub>, g-C<sub>3</sub>N<sub>4</sub> thin sheets and HS g-C<sub>3</sub>N<sub>4</sub>-O samples are estimated to be 1.117, 1.175



**Fig. 6.** (a) UV-vis absorption spectra of bulk g-C<sub>3</sub>N<sub>4</sub>, HS g-C<sub>3</sub>N<sub>4</sub>-O and g-C<sub>3</sub>N<sub>4</sub> thin sheets. (b) The corresponding bandgap estimated by related curves of  $(\alpha h\nu)^2$  vs photon energy plotted. (c) Mott-Schottky plots of bulk g-C<sub>3</sub>N<sub>4</sub>, HS g-C<sub>3</sub>N<sub>4</sub>-O and g-C<sub>3</sub>N<sub>4</sub> thin sheets. (d) Comparison of electronic band structure of bulk g-C<sub>3</sub>N<sub>4</sub>, HS g-C<sub>3</sub>N<sub>4</sub>-O and g-C<sub>3</sub>N<sub>4</sub> thin sheets.



**Fig. 7.** (a) Photocurrent responses of bulk g-C<sub>3</sub>N<sub>4</sub> and HS g-C<sub>3</sub>N<sub>4</sub>-O electrodes under visible light irradiation at 0.2 V vs Ag/AgCl in a 0.5 M Na<sub>2</sub>SO<sub>4</sub> aqueous solution. (b) PL spectra of bulk g-C<sub>3</sub>N<sub>4</sub>, HS g-C<sub>3</sub>N<sub>4</sub>-O and g-C<sub>3</sub>N<sub>4</sub> thin sheets by excited at 420 nm. (For interpretation of the references to color in the text, the reader is referred to the web version of this article.)

and 0.83 V, respectively. The  $V_{fb}$  could be converted to standard hydrogen electrode (SHE) by following the Eqs. (1)–(3):

$$E \text{ (V vs RHE)} = V_{fb} + E_{Ag/AgCl}^0 + 0.059 \times pH \quad (1)$$

$$E \text{ (V vs SHE)} = E \text{ (V vs RHE)} + 0.0592 \times pH \quad (2)$$

$$E \text{ (eV)} = -4.6 - E \text{ (V vs SHE)} \quad (3)$$

In the electron energy diagram with the unit eV, the vacuum energy is usually set to 0 eV, as a result, the  $V_{fb}$  vs SHE bulk g-C<sub>3</sub>N<sub>4</sub>, g-C<sub>3</sub>N<sub>4</sub> thin sheets and HS g-C<sub>3</sub>N<sub>4</sub>-O samples are calculated to be −3.68, −3.62 and −3.98 eV, respectively. Assuming the difference between  $V_{fb}$  and CB minimum is negligible for n-type semiconductors [29], the determined  $V_{fb}$  can be approximated as the CB edge ( $E_{cb} \approx V_{fb}$ ). Therefore, according to their band gap in Fig. 6b,

these valance band maximum (VBM) was approximatively at −6.3, −6.41 and −6.41 eV respectively. Fig. 6d shows schematic energy level diagrams of the bulk g-C<sub>3</sub>N<sub>4</sub>, g-C<sub>3</sub>N<sub>4</sub> thin sheets and HS g-C<sub>3</sub>N<sub>4</sub>-O in comparison with the potentials for water reduction and oxidation. It could be clearly seen that the VBM of HS g-C<sub>3</sub>N<sub>4</sub>-O has a similar position as that of g-C<sub>3</sub>N<sub>4</sub> thin sheets, whereas the CBM of HS g-C<sub>3</sub>N<sub>4</sub>-O is 0.36 eV lower than that of g-C<sub>3</sub>N<sub>4</sub> thin sheets. Therefore, it could be reasonably inferred that the oxygen edge doping significantly hybridizes atomic states by either the substitution of N sites by O atoms [10,19,30], or structural distortion by the introduction of guest oxygen atoms [31]. As mentioned before, in addition to the light absorptivity, the separation and transfer of photo induced charge carriers act also important role on its photocatalytic activity. Photo luminescence (PL) spectra and photo

current response were therefore used to investigate the separation and transfer of photo induced charge carriers. As shown in Fig. 7a, the photoresponses for both bulk g-C<sub>3</sub>N<sub>4</sub> and HS g-C<sub>3</sub>N<sub>4</sub>-O were reproducible during the repeated on/off cycles under visible light irradiation, suggesting efficient visible light harvesting. Moreover, the photocurrent of the HS g-C<sub>3</sub>N<sub>4</sub>-O electrode was about five times higher than that of the bulk g-C<sub>3</sub>N<sub>4</sub> electrode, indicating a much more enhanced separation efficiency of photoinduced electrons and holes. Fig. 7b shows the PL spectra of the bulk g-C<sub>3</sub>N<sub>4</sub>, g-C<sub>3</sub>N<sub>4</sub> thin sheets and HS g-C<sub>3</sub>N<sub>4</sub>-O excited by 420 nm at room temperature. The emission peak of the bulk g-C<sub>3</sub>N<sub>4</sub> appears at ca. 446 nm, attributed to the band–band recombination of the charge carriers with emission photon energy equal to its bandgap energy. Significantly, the blue shift and raising intensity of emission peak for g-C<sub>3</sub>N<sub>4</sub> thin sheets was due to shifting the conduction and valence band edges in opposite directions. Obviously, after oxygen edge doping, the emission peak was red-shifted with relative low intensity, suggesting that the recombination of electron–hole pairs might be effectively inhibited. According to above analyses and characterization, the improved photocatalytic activities for H<sub>2</sub> generation and Rh.B degradation could be ascribed to high surface area offering numerous catalytic sites, narrow band gap improving light harvesting and oxygen edge doping enabling more efficient charge generation and separation.

#### 4. Conclusion

In summary, a facile process was developed to obtain holey structured g-C<sub>3</sub>N<sub>4</sub> thin sheets with oxygen edge doping (HS g-C<sub>3</sub>N<sub>4</sub>-O) by photo-Fenton reaction. The holey structure efficiently increased surface area from 46 m<sup>2</sup> g<sup>−1</sup> of bulk g-C<sub>3</sub>N<sub>4</sub> to 348 m<sup>2</sup> g<sup>−1</sup>. The oxygen edge doping significantly improving light harvesting ability, and improves separation efficiency of the photoinduced charge carriers. As a results, the as-obtained HS g-C<sub>3</sub>N<sub>4</sub>-O presented high photocatalytic activities for H<sub>2</sub> evolution and Rh.B degradation. This study shows that this kind of noble-metal-free g-C<sub>3</sub>N<sub>4</sub> has a great potential for solar energy conversion.

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#### Appendix A. Supplementary data

Supplementary data associated with this article can be found, in the online version, at <http://dx.doi.org/10.1016/j.apcatb.2015.11.030>.

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